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² Original Research Paper

A facile synthesis of α -Ni(OH)₂-CNT composite films for supercapacitor $\frac{7}{5}$ application

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ABSTRACT

The α -Ni(OH)₂-CNT composite films have been successfully synthesized by a simple chemical method 29
and their supercapacitive properties were investigated by variation of CNT. The structural compositional 30 and their supercapacitive properties were investigated by variation of CNT. The structural, compositional, 30
morphological. wettability and electrochemical properties of the composite films were studied by using 31 morphological, wettability and electrochemical properties of the composite films were studied by using 31 various characterization techniques. X-ray diffraction analysis revealed that the synthesized composite 32 films are polycrystalline in nature. FT-Raman spectroscopy result showed the characteristic Raman band 33 of CNT and α -Ni(OH)₂ which confirmed the formation of α -Ni(OH)₂-CNT composite. SEM micrographs 34
showed porous microstructure of the synthesized films and hydrophilic nature of the films was con-
35 showed porous microstructure of the synthesized films and hydrophilic nature of the films was confirmed from wettability studies. Furthermore, the effect of the variation of CNT on the electrochemical 36 properties of the synthesized composite films was discussed. The electrochemical performance of the 37 composite films was studied by using cyclic voltammetry (CV) and Galvanostatic charge–discharge 38 (GCD) techniques. The α -Ni(OH)₂-CNT composite showed highest specific capacitance of 544 F g⁻¹ with 39 high retention capability of 85% after 1500th cycle and excellent cycling stability. high retention capability of 85% after 1500th cycle and excellent cycling stability.

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46 1. Introduction

47 Electrochemical supercapacitor has raised a great attention for 48 high power energy storage applications because of their excellent 49 cycling strength and high power density $[1,2]$. Also, it store energy 50 in the form of electric charges which are environmentally friendly. 51 Depending upon the charge storage mechanism, supercapacitor 52 can be divided into two type's viz electrochemical double layer 53 capacitor (EDLC) and pseudocapacitor [\[1,3\]](#page-7-0). However, pseudoca-54 pacitor has been extensively focused as compared to EDLC owing 55 to their high specific capacitance $[2]$. The various pseudocapacitive 56 materials, such as transition metal oxides/hydroxides like $RuO₂$, 57 MnO₂, NiO, Ni(OH)₂, and Co(OH)₂ has been successfully used as 58 electrode materials for supercapacitor applications $[4-8]$. Among 59 these materials, $Ni(OH)_2$ has been widely studied for supercapaci-60 tor applications by reason of its high theoretical specific capaci-61 tance, relatively good chemical stability and environmentally

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friendly nature [\[9\].](#page-7-0) But still have limitations in practical applica- 62 tions because of its low energy density (ED) and low electrical con- 63 ductivity. By considering these limitations, the development of 64 new advanced composite electrode material for supercapacitor 65 applications is a need of an hour. Recently, carbon based materials 66 are extensively used for electrochemical capacitor applications to 67 enhance their supercapacitive performance. It ranges from acti-
68 vated carbon (AC) to carbon nanotube (CNT) $[10-12]$. Especially, 69 CNT has been found to be an excellent form of carbon over the 70 other carbonaceous materials and have attracted an extensive 71 attention to prepare composite with metal oxides/hydroxides for 72 supercapacitor electrodes because of its high electrochemically 73 accessible surface area, excellent chemical stability and low resis- 74 tivity $[13,14]$. Additionally, CNT is used to alter the physical prop- 75 erties of metal oxides that are used for various applications such as 76 solar cells $[15]$, electrochromism $[16]$, electrochemical sensors $[10]$ 77 and supercapacitors [\[17\],](#page-7-0) respectively. The matrix of the state of \sim 78

M. Kazazi et al. [\[18\]](#page-7-0) have prepared nanoflakes nickel oxide-

79 carbon nanotube (NiO-CNT) composite thin films by elec- 80 trophoretic deposition for high-performance pseudocapacitor 81 applications. They have reported that the prepared NiO-CNT com- 82 posite electrode exhibited excellent pseudocapacitive behavior 83

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84 with a high specific capacitance of 786 F g^{-1} and 89.8% of capaci- tance retention after 1000th cycles as compared to pure NiO elec- trode. They have concluded that the uniformly dispersed CNT in the electrode that can provide fast and easy conductive pathway for transport of electrons into the active area of electrode material. 89 Cheng et al. [\[17\]](#page-7-0) have facile synthesized $Ni(OH)_2/CNTs$ nanoflake composites, which showed improved performance for supercapac- $\hspace{0.5cm}$ itor applications and gives a specific capacitance of 720 F g $^{-1}$. They have reported the better progress in electrochemical performance due to the synergetic effect of hydroxides and CNTs. Dai et al. [\[19\]](#page-7-0) have synthesized CNT-NiO nanocomposite by chemical con- version route and exhibited the high specific capacitance of 759 F g^{-1} in 6 M KOH electrolyte due to the synergistic effects.

 In the present manuscript, a simple chemical bath deposition (CBD) technique is used for the preparation of nickel hydroxide powder. After that an easy and cost-effective doctor blade method 100 is used for the synthesis of α -Ni(OH)₂-CNT composite films and their supercapacitive properties have reported. Such an easily syn- thesized electrode gives the high specific capacitance with remark- able rate ability and excellent cycling strength in 1 M KOH electrolyte. Also, the effect of CNT variation on the electrochemical 105 properties of the as-prepared α -Ni(OH)₂-CNT composite films is studied systematically and reported.

107 2. Experimental details

108 2.1. Synthesis of α -Ni(OH)₂-CNT composite films

 The chemicals were of analytical grade (AR) used without fur- ther purification. Multi-walled carbon nanotube (MWCNT) was purchased from Monad Nanotech Pvt. Ltd., Mumbai, and used without further purification. For preparation of the precursor solu-tion, double-distilled water (DDW) was used. Scheme 1 represents the steps involved for the synthesis of flower like microstructure α - 114 $Ni(OH)_{2}$ -CNT composite film. Initially, the nickel hydroxide powder 115 was prepared by using chemical bath deposition (CBD) method. In 116 this process, nickel sulfate (NiSO₄·6H₂O) as a nickel precursor, 117 potassium persulfate (K₂S₂O₈) as an oxidizing agent and aqueous 118 potassium persulfate ($K_2S_2O_8$) as an oxidizing agent and aqueous ammonia as a complexing agent were used. In particular, 20 ml 119 nickel sulfate (0.6 M) solution and 15 ml potassium persulfate 120 (0.25 M) solution were mixed in a 100 ml beaker and stirred well 121 at room temperature till its absolute dissolution and then get green 122 colored solution. Further, aqueous ammonia was added slowly into 123 the stirring solution to get pH 10.8 of the solution. Afterward, the 124 brown color precipitating solution was formed in a 100 ml beaker. 125 This obtained precipitate was filtered by using Whatman filter 126 paper and again washed several times with double distilled water 127 to remove the unwanted impurities. Finally, this washed precipi- 128 tate was dried for 24 h at room temperature. Following this proce- 129 dure the brown color nickel hydroxide powder was collected. 130

Further, the mixture of 0.5 g as-synthesized nickel hydroxide 131 powder, 0.05 g polyvinilidene fluoride, 0.025 g CNT and a small 132 amount of N-methyl-2-pyrrolidone was ground in agate mortar till 133 the paste was formed. This paste was deposited on stainless steel 134 substrate by doctor blade method followed by annealing at 135 300 °C for 1 h in an ambient atmosphere to remove the binders. 136 The resulting product α -Ni(OH)₂-CNT was denoted as composite 137 film NC-1. The above procedure was repeated by addition of 138 0.05 g, 0.075 g and 0.1 g CNT and these synthesized films were 139 denoted as composite films NC-2, NC-3 and NC-4, respectively. 140

2.2. Materials characterization 141

The identification of phase and crystalline structure of compos- 142 ite films were characterized by D2 PHASER, Bruker, X-ray diffrac- 143 tometer with Cu-K_{α} radiation (λ = 1.5406 Å) over 10°-90°. FT- 144

Scheme 1. Schematic of the steps involved for the synthesis of flowerlike porous microstructure α -Ni(OH)₂-CNT composite film.

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 Raman spectrum was recorded by using Bruker Multi RAM, Ger-146 many over 200–2000 cm^{-1} excited with the Argon 488 nm laser source obtained at room temperature. Surface morphological anal- ysis of the synthesized composite films was carried out by using JEOL JSM-6360 Japan made scanning electron microscopy (SEM). The identification of the elements onto the surface of composite film was analyzed by energy dispersive X-ray analysis (EDX) con- nected with FE-SEM instrument (FE-SEM, TESCAN). Wettability analysis of the composite films was studied by using Holmarc's contact angle meter (model no: HO-IAD-CAM-01). Electrochemical measurements (Cyclic Voltammetry, galvonostatic charge dis- charge and electrochemical impedance spectroscopy analysis) were carried out in 1 M KOH electrolyte by using Metrohm's Auto- lab 320 N with three-electrode cell method, wherein graphite and the saturated calomel electrode (SCE) were used as a counter and 160 the reference electrode, respectively. The composite film α -Ni 161 (OH)₂-CNT was prepared by doctor blade method and used as working electrode.

163 3. Results and discussion

164 3.1. XRD studies

165 Fig. 1(a–d) shows the XRD patterns of NC-1, NC-2, NC-3 and NC-166 4 composite films respectively. It clearly shows that all the films 167 are polycrystalline in nature. The diffraction peaks are located at 168 12.2°, 33.5°, and 59.6°, along the (003), (101) and (110) planes 169 respectively, which clearly corroborates the structure of pure α -170 Ni $(OH)_2$ phase [\[18\]](#page-7-0). Furthermore, the observed and calculated 'd' 171 values of diffraction peaks for α -Ni(OH)₂ are matched well with 172 standard ICPDS card no. 38-0715. Also, it is seen that, there is a prestandard JCPDS card no. 38-0715. Also, it is seen that, there is a pre-173 dominant diffraction peak located at 26.1° along the $(0\ 0\ 2)$ plane, 174 which confirms the formation of CNT. The corresponding 'd' value 175 of diffraction peak for CNT is matched well with standard JCPDS 176 card No. 75-1621 with hexagonal phase [\[18,20\].](#page-7-0) The above XRD 177 results clearly confirms the formation of α -Ni(OH)₂-CNT compos-178 ite. Also, in XRD pattern the stainless steel peaks are observed at 179 43.14° , 44.07 $^\circ$, 50.27 $^\circ$ and 74.20 $^\circ$, respectively which are indexed 180 by symbol *. The crystallite size of the composite films is calculated 181 by using the Scherrer's relation [\[21\].](#page-7-0)

Fig. 1. X-ray diffraction patterns of $Ni(OH)_2$ -CNT composite films: (a) NC-1, (b) NC-2, (c) NC-3 and (d) NC-4.

$$
D = \frac{0.9\lambda}{\beta \cos \theta} \tag{1}
$$

where D is the crystallite size in nm, λ is an incident X-ray 185
velepoth of Cu K_n line B is the full-width half-maximum 186 wavelength of Cu K_{α} line, β is the full-width half-maximum 186
(EWHM) of the diffraction neak and θ is the Bragg's diffraction 187 (FWHM) of the diffraction peak and θ is the Bragg's diffraction 187
angle in degree. The calculated value of crystallite size for all the 188 angle in degree. The calculated value of crystallite size for all the composite films is found to be in the range of 30–36 nm. Jahromi 189 et al reported that the influence of nickel oxide nanoparticles on 190 the supercapacitive performance [\[22\]](#page-7-0). 191

3.2. Raman spectroscopic analysis 192

The Raman spectrum of NC-4 composite film is shown in Fig. 2. 193 Three distinct peaks are observed at 503, 1331 and 1586 cm^{-1} . 194 Raman peak at 503 cm^{-1} corresponds to vibrational lattice mode 195 of Ni-OH and confirms the formation of α -Ni(OH)₂ [\[20\].](#page-7-0) The two 196 bands at 1331 cm⁻¹ (D band) and 1586 cm⁻¹ (G band) reveal the 197 typical CNT bands $[23,24]$. Raman band at 1586 cm⁻¹ originates 198 from Raman active in-plane atomic displacement E_{2g} mode [\[23\].](#page-7-0) 199 The Raman profile shows the formation of α -Ni(OH)₂-CNT compos- 200 ite film without impurities. The results analyzed from Raman anal- 201 ysis are well consistent with XRD analysis. 202

3.3. Morphological studies and energy dispersive X-ray (EDX) analysis 203

[Fig. 3](#page-3-0)(a–d) represents the SEM micrographs of all synthesized 204 α -Ni(OH)₂-CNT composite films such as NC-1, NC-2, NC-3 and 205 NC-4, respectively. In synthesis process, the mixture of $Ni(OH)_{2}$ 206 and CNT was ground in agate mortar to form strong adhesion 207 between them and also the quantity of $Ni(OH)_2$ was more than 208 that of CNT in the composite. Consequently the CNT merged into 209 the $Ni(OH)_2$ surface and formed together as flower-like porous 210 microstructures as seen in [Fig. 3](#page-3-0) (b-d), suggesting that the struc-
211 tural interaction as well as porosity of the composite film is pre- 212 served well after addition of CNT [\[25,26\].](#page-7-0) Moreover it is seen 213 that, these microstructures are getting agglomerated well because 214 of strong interactions between Ni(OH)₂ and CNT due to grinding, 215 and formed the abundant hollow spacing in the composite, which 216 is increased gradually with addition of CNT $[25,27]$. This unique 217 morphological surface provides more electrochemically active sites 218 to increase the interfacial contact area between electrode surface 219 and electrolyte for an easy ionic transportation during charging- 220 discharging processes, and consequently enhances the electro- 221

Fig. 2. Raman spectrum of α -Ni(OH)₂-CNT composite film (NC-4 composite film).

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Fig. 3. SEM images of α -Ni(OH)₂-CNT composite films: (a) NC-1, (b) NC-2, (c) NC-3 and (d) NC-4.

 chemical properties of the material, which is beneficial for super-223 capacitor application $[18]$. [Fig. 4](#page-4-0) illustrates that the typical EDX spectrum of the synthesized composite film NC-4. The prominent peaks of Ni, O and C elements are clearly visible in EDX spectrum 226 which confirms the synthesized film is composite of $Ni(OH)_2$ and CNT. The compositional values in atomic percentage of Ni, O and C elements are observed as 14.73, 53.71 and 31.56, respectively.

229 3.4. Wettability studies

230 Wettability analysis of the composite film is empirically evalu-231 ated by the measurement of contact angle (θ) between water dro-232 plet and the surface of the film. The water contact angle value is 233 directly related to the chemical composition, presence of local 234 inhomogeneity's and geometrical surface of the films. A water con-235 tact angle (θ) should be less than 90 \degree consisting of hydrophilic sur-236 face of the material. Fig. $5(a-d)$ represents the water contact angle 237 images of all synthesized composite films NC-1, NC-2, NC-3 and 238 NC-4 respectively. The measured contact angle values of all the 239 films indicate hydrophilic surface which is more important param-240 eter for good performance of electrode material. The SEM micro-241 graph of all composite films shows the presence of abundant 242 hollow spacing around the porous microstructures. A water droplet 243 placed on this unique surface is absorbed by these hollow spacing 244 and porous microstructures and then improve the hydrophilic sur-245 face of the composite. Also, the hydroxide group present in the α -246 $Ni(OH)₂-CNT$ composite may help the absorption of ions to 247 improve the hydrophilicity of the CNT $[2]$. So the contact angle of 248 water drop on α -Ni(OH)₂-CNT composite film decreases with addi-249 tion of CNT. Shaikh et al. $[28]$ reported that the surface of CPCNT 250 films transform from hydrophobic to hydrophilic (120 to 65°) with 251 annealing temperature. Yu et al. $[29]$ reported that the contact 252 angle of 16.3° was obtained with the 7 wt% of CNT content. Ouyang 253 et al. [\[30\]](#page-7-0) reported that the Ni-Ag/SWCNTs/GCE composite electrode exhibited the best hydrophilicity due to the smallest contact 254 angle of 30.8° . In the present case, the composite film NC-4 gives 255 lower contact angle value of 57° containing superior hydrophilic 256 surface, which provides the high effective surface area that can 257 improve conductivity of the material by reducing the effective 258 resistances during ionic exchanging process at electrode- 259 electrolyte interface. The hydrophilic behavior of the film permits 260 an easier access for redox reactions which is beneficial for superca- 261 pacitor applications. Also, such an important parameter plays an 262 effective role for reducing diffusion path length of the ions during 263 charge-discharge processes [\[31\]](#page-7-0). 264

3.5. Cyclic voltammetry (CV) studies 265

By CV and GCD measurements the capacitive nature of the 266 material can be examined. Fig. $6(a-d)$ shows the cyclic voltammo- 267 grams (CV) curves of all α -Ni(OH)₂-CNT composite films i.e. NC-1, 268 NC-2, NC-3 and NC-4, respectively at different scan rates from 10 269 to 100 mVs^{-1} . The CV curves were recorded in the potential win-
270 dow of -1.0 V to 0.4 V vs SCE in 1 M KOH electrolyte. It is seen that, 271
the recorded CV curves of all composite films are nearly identical 272 the recorded CV curves of all composite films are nearly identical and exhibit a quasi rectangular shape with an oxidation and reduc- 273 tion peaks respectively, which implies an ideal pseudocapacitive 274 characteristic of the material $[32]$. Also, the charge-discharge pro- 275 cesses of the films are associated with an oxidation and reduction 276 pair. The corresponding faradaic redox reaction in which anodic 277 peak at 0.3 V vs SCE correspond to the oxidation of α -Ni(OH)₂ to 278 γ -NiOOH, whereas the cathodic peak at -0.5 V vs SCE correspond 279
to the reverse process is given as follows [32]: 280 to the reverse process is given as follows $[32]$: 281

 α -Ni(OH)₂ + OH⁻ = γ -NiOOH + H₂O + e⁻ (2) 283

From Fig. $6(a-d)$ it is seen that, the area under the CV curve 284 increases with increasing the scan rate from 10 to 100 mVs^{-1} which 285 reveals that the voltammetric current is directly proportional to the 286

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Fig. 4. EDX spectrum of typical α -Ni(OH)₂-CNT composite film (NC-4 composite film).

Fig. 5. Water contact angle images of α -Ni(OH)₂-CNT composite films: (a) NC-1, (b) NC-2, (c) NC-3 and (d) NC-4.

 scan rate [\[33\]](#page-7-0). Moreover, the current density of both anodic and cathodic peaks smoothly increases by increasing the scan rate but, there is no any apparent changes in the shape of the CV curves which, signify lowest contact resistance of the electrode for better electrochemical processes [\[34\].](#page-7-0) Nevertheless, the specific capaci- tance of the composite film decreases with increasing the scan rate because of moderated reaction mechanism between electrode-294 electrolyte interfaces. [Fig. 7](#page-5-0) shows the collectively depicted CV curve of all synthesized composite films NC-1, NC-2, NC-3 and 296 NC-4, respectively measured at a constant scan rate of 10 mVs $^{-1}$ in 1 M KOH electrolyte. It indicates that an area under the CV curve of the composite films increases with increasing the quantity of the CNT. The CV curve area is used to calculate the specific capacitance of the composite film by using the following equation [\[35\]](#page-7-0), and pre-sented in [Table 1](#page-5-0). ³⁰²

$$
C_s = \frac{\int i dv}{2(m)(\Delta V)(Vs)}
$$
(3)

305 where C_s is the specific capacitance, \int i dv is the area of the CV 306 curves within the assigned potential range, m is the mass of the 307 active material on a substrate, ΔV is the potential window, and Vs is the scan rate. It is seen that the composite film NC-4 acquire high-
308 est area under the CV curve which provides the highest specific 309 capacitance of 544 F g^{-1} as compared to other composite films. 310 Hence, it corroborates, the role of CNT is important to increase 311 the specific capacitance of the composite film. The NC-4 sample 312 exhibits lowest water contact angle and maximum surface area 313 which offer the large number of electrochemical active sites for 314 the intercalation of ions from the electrolyte to the surface of the 315 electrode and hence faster diffusion into the active electrode mate- 316 rial takes place. 317

3.6. Galvanostatic charge-discharge (GCD) studies 318

The GCD study is very important for resolving the charge- 319 discharge stability, energy density and power density of the mate- 320 rial. [Fig. 8\(](#page-6-0)a-d) shows the charge-discharge curves of all α -Ni 321 $(OH)_2$ -CNT composite films i.e. NC-1, NC-2, NC-3 and NC-4, respec- 322 tively at constant current density of 1 mA/cm^2 were recorded in 323 the potential window of -1.0 V to 0.4 V vs SCE in 1 M KOH elec-
trolvte. All the GCD curves are seen as non-triangular in shape 325 trolyte. All the GCD curves are seen as non-triangular in shape which confirms the faradaic (pseudocapacitive) behavior of the 326

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Fig. 6. Cyclic voltammograms of α -Ni(OH)₂-CNT composite films: (a) NC-1, (b) NC-2, (c) NC-3 and (d) NC-4 in 1 M KOH electrolyte at various scan rates.

Fig. 7. Cyclic voltammograms of α -Ni(OH)₂-CNT composite films: (a) NC-1, (b) NC-2, (c) NC-3 and (d) NC-4 in 1 M KOH electrolyte at constant scan rate of 10 mV/s.

electrode material [\[36\]](#page-7-0). From the slope of GCD curves, it is seen 327 that the composite film NC-4 shows a longer discharging time 328 and gives maximum specific capacitance of 537 F g^{-1} at 1 mA/ 329 cm^2 . The specific capacitance (C_s), energy density (E) and power 330 density (P) of the composite films is calculated from discharge 331 curves, according to the following equation [\[37\],](#page-7-0) 332 333

$$
C_s = \frac{I}{m(\Delta V/\Delta t)}
$$
(4) 335

$$
E = \frac{1}{2}C(\Delta V)^2 \tag{5}
$$

339

336

$$
P = \frac{E}{\Delta t} \tag{6}
$$

where, $I -$ applied current, $m -$ mass of the deposited material onto 342 the electrode, ΔV – discharge voltage range and Δt – discharge 343 time. The calculated values of specific capacitance, energy density 344 and power density of composite films are presented in Table 1. 345 From Table 1 it clear that, the specific capacitance of α -Ni(OH)₂- 346

Fig. 8. Galvanostatic charge discharge curves of α -Ni(OH)₂-CNT composite films: (a) NC-1, (b) NC-2, (c) NC-3 and (d) NC-4 in 1 M KOH electrolyte at constant current density of 1 mA/cm² .

 CNT composite films are improved due to the contribution of con- ductive nature of CNT, which provides higher accessible active sites for fast redox process. The stability test measurement is carried out for optimized NC-4 composite film in 1 M KOH electrolyte and the recorded curves are shown in Fig. 9. It is seen that the capacitive retention varies with the cycle number. The 85% capacitance reten- tion is observed after 1500th cycle for typical composite film NC-4. 354 The stability test indicates that, the composite film α -Ni(OH)₂-CNT attains good cycling capacity in 1 M KOH electrolyte and hence it is a powerful electrode material for electrochemical mechanism. Also, 357 from Fig. 9 it is seen that, only 13% capacitive loss is occurred in an initial capacitance after 1000th cycle which reveal the remarkable 359 rate capacity of α -Ni(OH)₂-CNT composite electrode.

360 3.7. Electrochemical impedance spectroscopy (EIS) studies

361 The EIS studies are being performed to understand the reaction 362 kinetics of the material and to evaluate their overall resistance

Fig. 9. Variation of specific capacitance $(\%)$ with respect to cycle numbers for α -Ni (OH)2-CNT composite film (NC-4 composite film) in 1 M KOH electrolyte at a scan rate of 100 mV/s for 1500 cycles.

Fig. 10. Nyquist plot of α -Ni $(OH)_2$ -CNT composite films: (a) NC-1, (b) NC-2, (c) NC-3 and (d) NC-4 in 1 M KOH electrolyte.

components. In EIS analysis the larger diameter semicircle 363 observed in the high frequency region is related to the higher 364 charge transfer resistance (R_{ct}) of the electrode-electrolyte inter- 365 face and an inclined straight line behavior observed in the low fre-
366 quency region which corresponds to the limiting ion diffusion 367 process [\[38\].](#page-7-0) Fig. 10(a–d) shows the Nyquist plot of α -Ni(OH)₂- 368
CNT composite films NC-1, NC-2, NC-3 and NC-4, respectively in 369 CNT composite films NC-1, NC-2, NC-3 and NC-4, respectively in the frequency range from 10 mHz to 1 MHz for 1 M KOH elec- 370 trolyte. The EIS analysis exhibited that, the decrement in R_{ct} value 371 of composite films which are given in [Table 1.](#page-5-0) It indicates that the 372 improvement in conductivity of the electrodes due to the addition 373 of CNT in the composite $[26]$. It is observed that, the composite film 374 NC-4 exhibits smaller semicircle provides lowest charge transfer 375 resistance of the electrode-electrolyte interface and the higher 376 slope offer the minimum ion diffusion resistance. Thus, the com-
377 posite film NC-4 gives a very low R_{ct} value 3.32 Ω implying its high 378 conductivity and high capacitance as compared to the other com- 379 posite films [\[34,39\]](#page-7-0). 380

4. Conclusions 381

A flowerlike porous microstructure of α -Ni(OH)₂-CNT compos- 382 ite films have been synthesized by a simple and inexpensive doctor 383 blade method. The composite films showed excellent supercapac- 384 itor properties in 1 M KOH electrolyte. X-ray diffraction analysis 385 showed that the composite films are polycrystalline in nature. 386 FT-Raman analysis confirms the well formation of α -Ni(OH)₂-CNT 387 composite. Morphological studies shows porous microstructure 388 of the synthesized films and the wettability studies shows superior 389 hydrophilic surface of the composite film (NC-4). The electrochem-
390 ical analysis showed that the α -Ni(OH)₂-CNT composite film (NC- 391) 4) gives highest specific capacitance of 544 F g^{-1} and higher reten- 392 tion capability of (85%) after 1500th cycle. Moreover, it exhibited 393 the high energy density of 21.25 Wh/kg and high power density 394 of 4.78 kW/kg. From EIS measurements it confirmed that the α - 395 Ni(OH)₂-CNT composite electrode provides very lowest charge 396 transfer resistance (3.32 Ω) of the electrode-electrolyte interface 397
implying higher conductivity of the material. Thus, it is concluded 398 implying higher conductivity of the material. Thus, it is concluded that, the α -Ni(OH)₂-CNT composite film (NC-4) is more appropri- 399 ate material for supercapacitor applications. 400

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